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Mesoporous GO-TiO₂ Nanocomposites for Flexible Solid-state Supercapacitor Applications.

Priyanka Rani,¹ Arup Ghorai,¹ Saptarsi Roy,² Dipak K Goswami,³ Anupam Midya,^{1*} Samit K Ray⁴

¹School of Nanoscience and Technology, Indian Institute of Technology Kharagpur, Kharagpur 721302, India,

²Department of Mechanical Engineering, Aryan Institute of Engineering and Technology, Bhubaneswar, 752050, India

³Department of Physics, Indian Institute of Technology Kharagpur, Kharagpur 721302, India

⁴S. N. Bose National Centre for Basic Sciences, Kolkata 700 106, India

Corresponding email: anupam.midya@iitkgp.ac.in; anumidya@gmail.com:

Abstract:

Fabrication of flexible solid-state supercapacitor is a field of paramount importance because of its potential application in portable devices. In this article, graphene oxide and titanium dioxide hybrid (GO-TiO₂) nanocomposites of different compositions are employed for the fabrication of flexible solid-state supercapacitors. An easy, fast, scalable and controlled sol-gel method is devised for GO-TiO₂ nanocomposites growth from different amounts of graphene oxide and titanium isopropoxide at atmospheric pressure. Highly monodispersed rutile TiO₂ nanospindle is homogeneously grafted on GO by a controlled HCl catalyzed reaction. BET N₂ adsorption-desorption isotherm analysis confirms formation of the mesoporous structure having a large specific surface area favourable for faster reversible adsorption and desorption of ions. An optimum composition of GO-TiO₂ nanocomposite (TG25) exhibits a high areal specific capacitance of 73.43 mF/cm² at a current density of 0.5 mA/cm² due to formation of electrical double layer in a solid-state supercapacitor. The fabricated device shows high power density (3.5 mW/cm²), a high energy density (0.007 mWh/cm²), good flexibility and excellent cycling stability, 92% specific capacitance retention after 10,000 continuous charge-discharge cycles. Finally, three supercapacitors in a series have illuminated a red LED, indicating the nanocomposite as a potential candidate for energy storage technology.

Keywords: Rutile TiO₂, GO, Nanospindles, Nanocomposites, gel electrolyte, Supercapacitors.

Introduction:

Solid-state supercapacitors having a high power density and high specific capacitance have attracted great attention for their high stability, flexibility, long term use and they are complementary of battery in energy storage applications.[1] The performance of electrical double layer (EDL) supercapacitors, store energy due to reversible charge accumulation and desorption, can be improved by engineering the critical parameters such as increasing the electrical conductivity, [2] surface area and interconnectivity of active electrode materials.[3] In this respect, various transition metal oxides such as MnO₂, Nb₂O₅, RuO₂, Co₃O₄, and Ni(OH)₂ have been employed in supercapacitor having different structure and morphology which attributed to high specific surface area. In addition, transition metal oxides shows large pseudocapacitance due to surface redox reactions.[4] RuO₂ is one of the most promising electrode materials due to its ultrahigh pseudocapacitance (700 F g⁻¹), low resistivity, high chemical and thermodynamic stability under electrochemical environments. However it's application is restricted due to the high cost.[5] Co₃O₄ has high theoretical capacitance (up to 3560 F g⁻¹), and well-defined redox behavior but delivers poor rate capability and reversibility due to its poor electrical conductivity and large crystallite size.[6] MnO₂ can also be used as electrode material because it is predicted to show a high capacitance and it is inexpensive. But its poor conductivity (10⁻⁵–10⁻⁶ S cm⁻¹) limits the charge/discharge rate for high-power applications.[7] TiO₂ forms conventional electrical double-layer capacitor (EDLC) and shows rectangular cyclic voltammetry curves which make it a promising electrode material for supercapacitor devices. TiO₂ nanostructures have tunable high surface area, low cost, environmentally benign behavior and long term chemical stability which make it attractive

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3 material in bio-implant and other sophisticated energy storage application.[8] However, it has a
4 lower theoretical capacity (1206 F g^{-1}) and its high resistivity results in large internal resistance,
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6 slow ion diffusion and poor electron transfer.[9]·[10] On the other hand, supercapacitors of the
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8 bare carbon material exhibit high surface area, good conductivity, and restricted ionic
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10 accessibility.[11] Therefore, it is interesting to pair transition metal oxides such as TiO_2 with
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12 carbonaceous materials such as graphene to form composites to extract the benefits of both
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14 materials reducing the limitation of each material.[12] Restacking of graphene sheets can be
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16 minimized by using TiO_2 nanoparticles as intercalator, which eventually increases the specific
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18 surface area, sustaining better access of ionic electrolyte. Graphene, a sp^2 -hybridized two-
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20 dimensional carbon sheet, exhibits extraordinary mechanical flexibility,[13] high chemical
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22 stability, high electrical conductivity, and high specific surface area ($2630 \text{ m}^2 \text{ g}^{-1}$) attractive for
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24 the flexible device applications.[14]·[15]
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31 Few graphene- TiO_2 nanocomposites have been synthesized at high pressure or using surfactants
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33 or additives to minimize the agglomeration of TiO_2 . [16]·[17] For example, Xiang et al.[18] have
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35 reported shape and coupling effects of hydrothermally synthesized reduced graphene oxide-
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37 TiO_2 composites for supercapacitor applications using expensive nickel disk as the current
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39 collector. A bare TiO_2 nanotube array is also employed in supercapacitor to achieve a low aerial
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41 capacitances of 2.6 mF cm^{-2} . [19] Size-tunable anatase TiO_2 nanospindles assembled with
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43 titanium oxynitride/titanium nitride graphene nanocomposites show high cycling performance in
44
45 rechargeable lithium ion batteries.²³ Flexible composite film of rGO/ TiO_2 , grown at high
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47 temperature ($600 \text{ }^\circ\text{C}$), is employed as supercapacitor electrode in three-electrode cell
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49 configuration.[20] However, controlling the size of particles, their morphology and dispersion
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51 density of inorganic nanocrystals on graphene sheets, which eventually increase the electrical
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3 conductivity and surface area leading to a superior supercapacitor performance in two electrode
4 configuration is still a challenge, and needs more attention.
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8 Here, we report an easy, controlled, in-situ growth of mesoporous GO-TiO₂-nanocomposites of
9 different compositions at atmospheric pressure. The evolutions of HCl catalyzed TiO₂
10 nanospindles of rutile phase on graphene platform are monitored by SEM, TEM, XPS, XRD and
11 Raman spectroscopy techniques. The nanocomposites are employed in flexible solid-state
12 supercapacitor device using acetylene black conducting filler and a gel PVA-H₃PO₄ electrolyte
13 on flexible steel plate electrode. The devices show excellent cyclic stability (92% specific
14 capacitance retention after 10,000 charge-discharge cycles), high specific capacitance and power
15 density, indicating the GO-TiO₂ nanocomposite as a potential candidate for energy storage
16 applications. The novelty of the work lies in the easy growth of mesoporous GO-TiO₂
17 nanocomposite and demonstration of flexible solid-state supercapacitor igniting LED bulbs.
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31 32 **Materials and Characterizations:**

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34 Graphite powder, KMnO₄, Sodium nitrate, Titanium tetraisopropoxide, 30% H₂O₂, and conc.
35 H₂SO₄ (98%), HCl were purchased from Sigma Aldrich and used without further purifications.
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37 Ultra pure water was obtained from Sartorius ultra pure water purification system (Model Arium
38 Comfort II) and ultra pure water was used for the whole experiment.
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46 **Physical Characterization**

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48 The morphologies of the prepared samples were characterized by Field emission scanning
49 electron microscopic (FESEM). FESEM Images were taken by SUPRA 40 Field Emission
50 Scanning Electron Microscope (Carl Zeiss SMT AG, Germany). High Resolution Transmission
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3 Electron Microscopy images were taken by JEOL 200 keV instrument. The phase analysis and
4 structure were investigated by X-ray diffraction measurements. Raman spectroscopy (Lab RAM
5 HR JovinYvon) is carried to study the crystalline quality and local structural disorder of
6 nanocomposites. The surface chemical composition and chemical states of the elements in the
7 samples were identified by X-ray photoelectron spectroscopy (model PHI 5000 Versa Probe II
8 (ULVAC, PHI Inc.). The electrochemical properties and capacitance measurements of the GO-
9 TiO₂ nanocomposites electrodes were characterized using cyclic voltammetry (CV),
10 galvanostatic charge discharge (GCD) and electron impedance spectroscopy (EIS) using an
11 electrochemical station (Gamry, Model 1000 interface) with a two-electrode configuration using
12 H₃PO₄/PVA electrolyte, GO-TiO₂ nanocomposite active material deposited on electrodes on steel
13 plate. The EIS measurements were conducted at varying frequency from 0.001 Hz to 10 kHz
14 with amplitude of 5 mV at an open-circuit voltage. The CV and GCD were tested between 0 and
15 1 V, and the areal specific capacitance (mF/cm²) was calculated from discharge curves from the
16 following equation.
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$$36 \quad C = \frac{(I\Delta t)}{(A\Delta V)}$$

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41 Where I , Δt , A and ΔV is discharged current, discharge time, area of electroactive material
42 and potential window, respectively.
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47 At a given scan rate the specific energy density (E.D; mWh/cm²) and Specific power density
48 (P.D; mW/cm²) of the device was calculated from the following equation
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$$52 \quad E.D = \frac{C\Delta V^2}{7200}$$

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$$P.D = \frac{E.D \times 3600}{\Delta t}$$

Δt is the duration (sec) of the discharge half-cycle in the discharge curve.

Synthesis of GO

Graphite oxide was synthesized by the modified Hummer and Offeman's method with graphite powder.[21] Graphite powder was oxidized using KMnO_4 in presence of conc H_2SO_4 . Graphite oxide slurry was filtered and exfoliated in DI water by ultrasonication in a bath sonicator as described in our previous work.[22] The graphene oxide was purified sequentially by centrifugation and dispersion of the precipitations in DI water. The GO was collected after removing small and big graphene oxide particle at a high and low centrifugation speed, respectively.

Synthesis of GO-TiO₂

The schematic of GO-TiO₂ nanocomposite synthesis is shown in Figure 1. Concisely, 0.937 μL of titanium tetraisopropoxide (TTIP) was dissolved in 40 ml ultrapure DI containing 5 ml HCl under vigorous stirring with a magnetic stirrer for 30 min. Different amounts of as-synthesized GO (10, 25, 50 and 100 mg) are added to the solution and stirred for 30 min, separately. After that, the temperature of the growth solution was maintained at 120°C for 2 hr for each sample. A gray precipitation was obtained. The resultant precipitation was washed with distilled water for three times by centrifugation at 10,000 rpm after dispersing and followed by discarding the supernatant to remove remaining ions in the by-product. The precipitation was collected and

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3 dried in an oven at 100 °C for 12 hr for further characterization and device fabrications. Few
4 control TiO₂ samples were also prepared without addition of GO at similar condition using
5 different acid concentrations (0.5 M HCl, 0.5 M HNO₃, 1M HCl). The resultant GO-TiO₂
6 nanocomposites were termed as TG10, TG25, TG50, and TG100 prepared using 10, 25, 50, and
7 100 mg of GO, respectively.
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15 **Supercapacitor device fabrication:**

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19 Electrodes comprised of GO-TiO₂ nanocomposite as active material was prepared by mixing the
20 85:15 mixture of GO-TiO₂ nanocomposite and acetylene black in N, N-dimethylformamide
21 (DMF). The solution of the active material was prepared by stirring the mixture of GO-TiO₂
22 nanocomposites and acetylene black for overnight at 60 °C in DMF. The black color solution
23 was drop casted on to stainless steel plates (0.01 mm width) and dried at 80 °C for 30 min in the
24 oven. The active layer film was soaked in gel electrolyte aq. PVA-H₃PO₄ gel electrolyte for 10
25 min. The gel electrolyte was prepared by dissolving 1g PVA in 10 ml DI water at 60 °C for 12
26 hr, then 1 ml H₃PO₄ was added into the solution and stirred for 2 hr. After drying in an oven for 2
27 hr at 60 °C, the electrodes were kept for 24 hr at room temperature. Next, the two symmetric
28 fabricated electrodes were sandwiched carefully. PVA plays a dual role here as a binder to hold
29 the active layer as well as a separator between two electrodes preventing shorting. To evaporate
30 extra amount of water, the device was further dried at 60 °C in an oven for 2 hr before the
31 measurement was performed.
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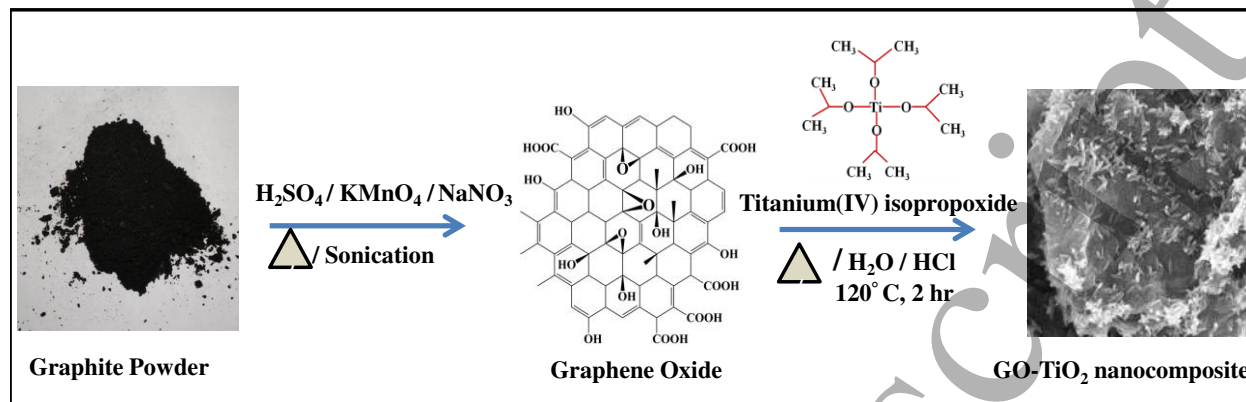


Figure 1: Schematic diagram of the evolution of TiO₂ nanospindles on graphene oxide.

Results and Discussions:

Figure 2(a) shows the FESEM image of agglomerated TiO₂ nanospindles in the absence of GO. A typical wrinkle structure of multilayer GO sheet is found in FESEM image, as shown in Figure 2(b). Whereas homogeneously dispersed nanospindles on GO is shown in Figure 2(c). Incorporation of different proportions of GO precursor to the same TTIP amount (0.1:1, 1:0.25:1, 1:0.5:1 and 1:1) dispersion density of TiO₂ nanospindles can be tailored on GO sheets as shown in Figure S1 (ESI). Compared with homogeneous growth in the solution, the nanospindles prefer to nucleate heterogeneously at the defect sites (oxygen containing moieties) on the GO sheets so free unattached single nanospindles can't be found in the product. There is no effect seen on morphology, including the size and shape by varying the reaction time after two hours, while keeping other conditions constant. Nanospindles are grafted uniformly onto the graphene oxide sheets after induction of GO to the TTIP precursor, keeping the nanospindle-shape of TiO₂ unchanged. However, the size of TiO₂ nanospindles changes from 154 to 311 nm in length and

46 to 118 nm in diameter, as shown in Table S1 (ESI) with the varying proportion of GO from 1:0.1 to 1:1.

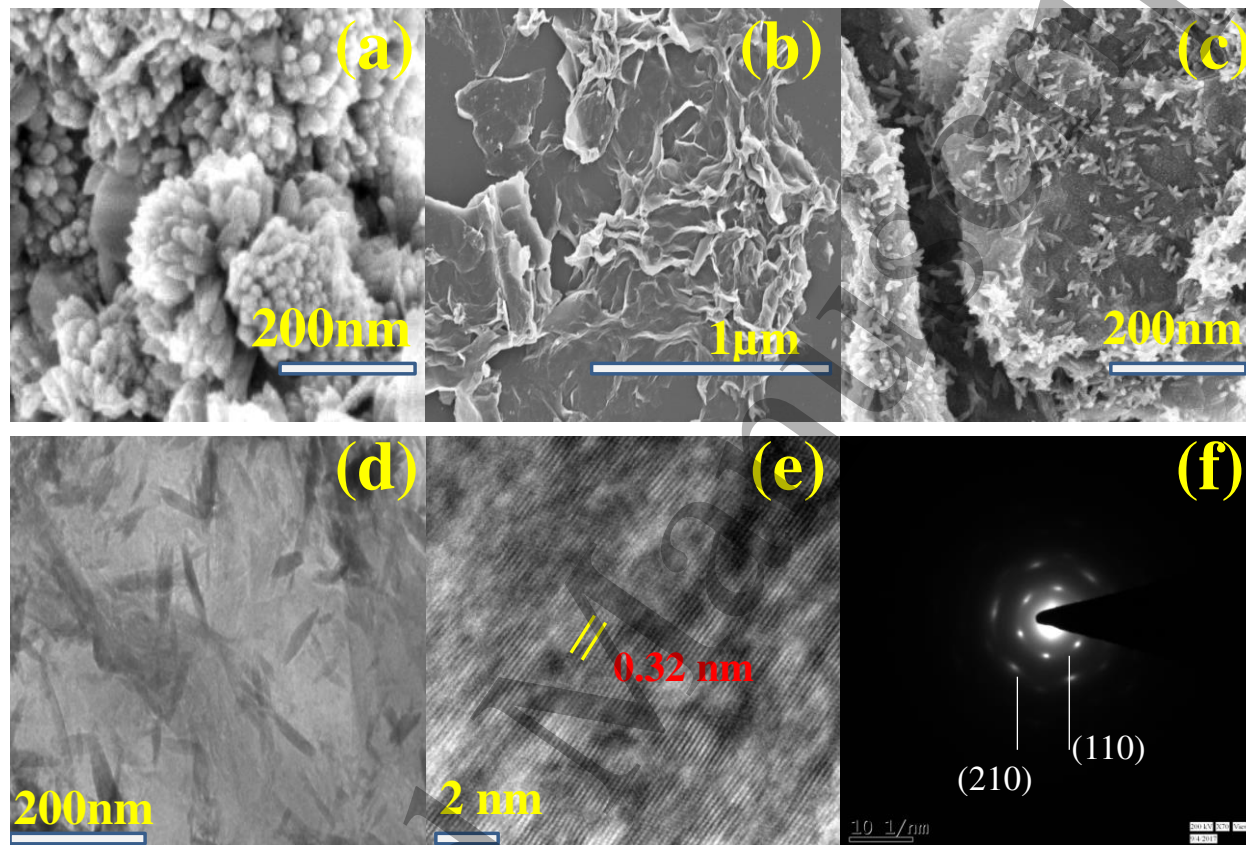


Figure 2: (a) Typical FESEM image of bare TiO_2 ; (b) graphene oxide; (c) TG 25. (d) Typical TEM image of TG25; (e) HRTEM image of TG25; (f) SAED pattern of TG25.

Typical FESEM image of TG25 is shown in Figure 2(c) where ~ 268 nm long nanospindle of ~ 78 nm diameter is homogeneously grafted on GO. Figure 2 (d) shows the TEM images of TiO_2 nanospindle grafted on graphene oxide in TG25. Presence of TiO_2 prevents the restacking of graphene oxide layers and expected to exhibit a higher specific surface area which is essential for supercapacitor applications.[23] It may be noted that after a prolong sonication and washing TiO_2 nanospindles are remain attached to GO sheets, indicating a strong interaction between GO sheets and TiO_2 nanospindles. This intimate hybridization endows easy and fast charge transport

between GO and TiO₂ nanospindles beneficial for improved supercapacitor performance. The lattice fringe spacing of TiO₂ nanoparticle in HRTEM image (Figure 2 (e)) is found to be 0.32 assigned to the (110) inter planner distance of the rutile phase.[24]·[25] Selected area electron diffraction (SAED) patterns confirm good crystallinity and the lattice planes of the pure tetragonal-rutile TiO₂ phase.

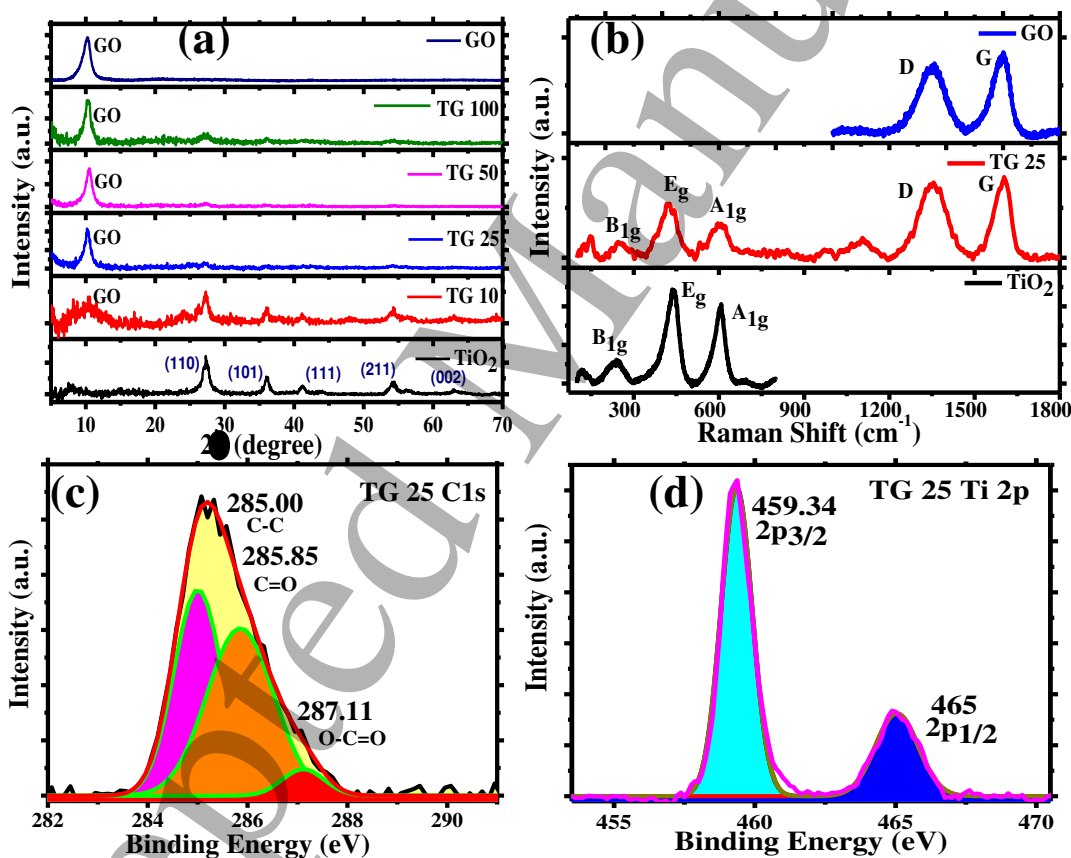


Figure 3: (a) Typical XRD patterns of bare TiO₂ and graphene oxide and TG25. (b) Raman spectra of different GO-TiO₂ nanocomposites. (c) X-ray photoelectron spectra of core level C1s electron of TG25. (d) Ti2p core level electron of TG25.

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3 The crystal structure of TiO₂, GO-TiO₂ nanocomposites, and graphene oxide were characterized
4 using X-Ray diffraction (XRD) analysis and the XRD patterns are shown in Figure 3(a). For
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6 TiO₂ and GO-TiO₂ nanocomposites, all of the peaks can be indexed as the tetragonal rutile phase
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8 with primitive lattice, space group- P42/mnm (136), and lattice constant $a = b = 4.601$ $c = 2.956$
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10 from (JCPDS 89-0552). Interestingly, it is observed that the concentration of hydrochloric acid is
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12 critical to obtain Rutile phase TiO₂. Below 1.0 M concentration of HCl produces a mixture of
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14 rutile and anatase. Whereas above 0.5 M concentration of nitric acid produces spherical particles
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16 of anatase phase as shown in Figure S2 (ESI). When the concentration of hydrochloric acid in
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18 the solution is higher than 1.0 M, the product is nanospindle of pure rutile phase, (Figure S3,
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20 ESI). Even at the same concentration, hydrochloric acid is more effective for the formation of the
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22 rutile phase than nitric acid. [26] From above observations, we can conclude that HCl plays
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24 major role in the reaction: (i) due to an acidic medium, hydrolysis rate of TTIP got suppressed
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26 and slowed down and (ii) HCl is the major factor for the formation of the nanospindle shape of
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28 TiO₂.
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36 The bonding structures of the GO-TiO₂ nanocomposites were characterized by Raman
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38 spectroscopy shown in Figure 3(b). The characteristic Raman peaks for TiO₂ and TG25
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40 nanocomposites are observed at ~ 441 and 606 cm^{-1} attributed to the E_g and A_{1g} modes of
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42 rutile TiO₂, respectively. Two characteristic peaks are observed for bare GO at ~ 1349 and
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44 1592 cm^{-1} , and for TG25 at ~ 1352 and 1597 cm^{-1} , corresponding to the well-defined D and
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46 G bands for the defect and graphitized structures, respectively. The D-band corresponds to the
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48 disorder with sp³ structural imperfections. Therefore, the intensity ratio (I_D/I_G) for GO is found to
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50 be 0.83, signifying the presence of the large amount of defect (sp³ C) in the sample. The change
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52 of I_D/I_G from 0.83 to 1.30 with the increasing proportion of TiO₂ indicates a strong hybridization
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of TiO₂ and graphene oxide. Raman spectra of GO and different GO-TiO₂ nanocomposites are shown in Figure S4.

The nanocomposite formation is also investigated by X-ray photoelectron spectroscopy (XPS). Figure 3(c) shows the C1s electron binding energy peaks, with a binding energy of 285 eV attributed to the C–C and C=C bonds, while the peaks centered at the binding energies of 285.85, 287.11eV correspond to carbonyl (C=O) and carboxyl (O-C=O) carbon, respectively. The presence of Ti2p core level photoelectron spectrum of Ti 2p_{3/2} at 459.34 eV and Ti 2p_{1/2} at 465 eV in Figure 3(d) is a signature of Ti (IV). Figure S5 (ESI) shows the comparative photoelectron spectra of Ti2p core level electron for different compositions of nanocomposite. The broadening of the peak for Ti2p core level electron with the increase in the proportion of GO in Figure S5 (ESI) indicates intimate mixing of TiO₂ and GO.

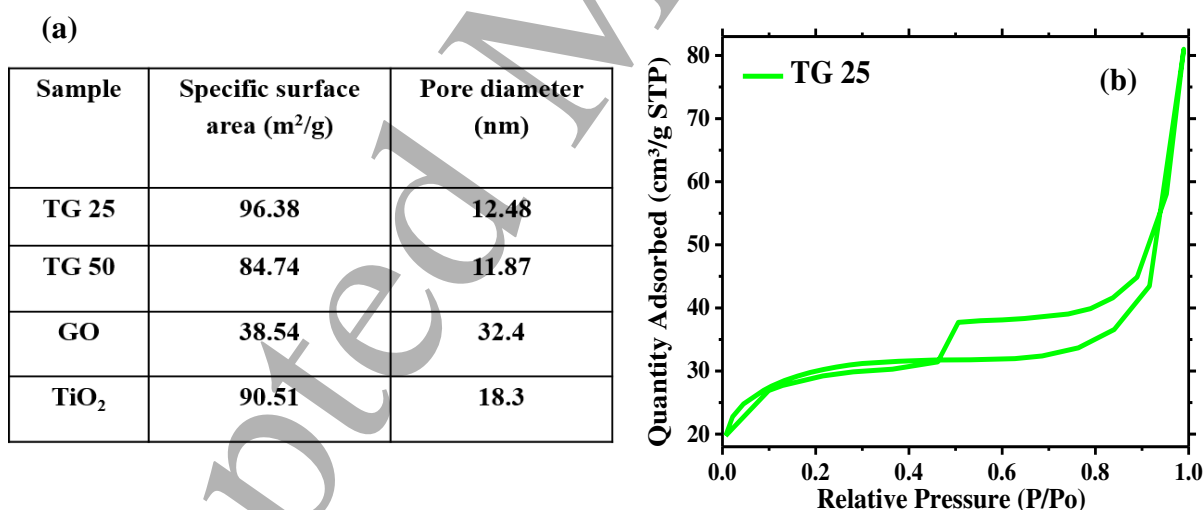


Figure 4: (a) Textural properties of GO-TiO₂ nanocomposites; (b) BET N₂ adsorption–desorption isotherm of TG25.

To get an insight into the surface property and supercapacitor performance of the nanocomposites, the specific surface area, a pore diameter of bare TiO₂, TG 25, TG50 and bare GO are investigated by BET nitrogen adsorption-desorption isotherm measurements

using Beckman Coulter SA3100 surface area analyzer. Specific surface area and pore diameter of nanocomposites are tabulated in (Figure 4a). TG 25 has shown higher surface area than TG50, bare TiO_2 and bare GO. The high specific surface area of TG25 nanocomposite may attribute to a superior specific capacitance performance in a supercapacitor. From BET measurement, the average pore diameter of TG25 is found to be 12.48 nm, indicating mesoporous nature of the electrode material which allows easy adsorption and desorption of smaller H^+ (1.2 Å) and PO_4^{3-} (2.38Å) ions and hence contributing to the capacitance of the supercapacitor. The TG25 shows type IV isotherm (Figure 4b) which is a characteristics of mesoporous structures.

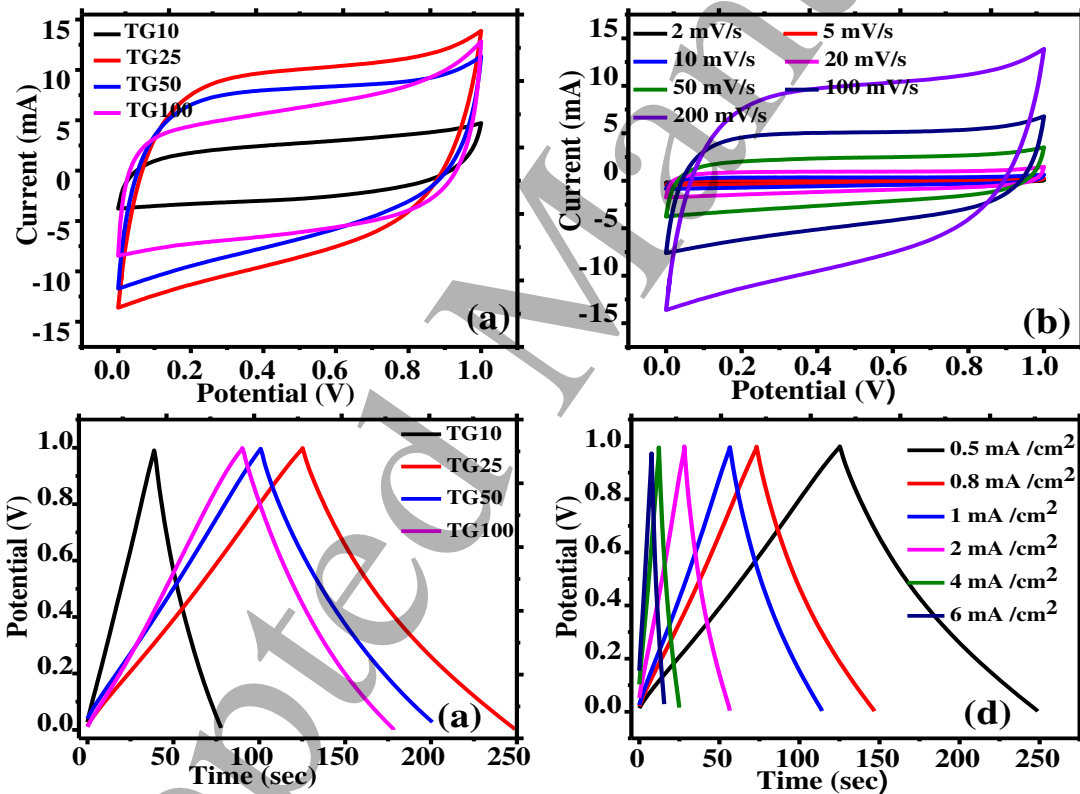


Figure 5: Typical CV curves of (a) different compositions of GO in TiO_2 at 200 mV/s scan rate; (b) TG 25 at different scan rates. Charge-discharge curves of (c) different compositions of GO- TiO_2 at 0.5 mA/cm² current density; (d) TG25 at different current densities.

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3 Electrochemical charge storage properties of the fabricated solid-state supercapacitor devices
4 were evaluated by cyclic voltammetry (CV). Figure 5(a) presents the steady state CV curves of
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6 GO-TiO₂ based solid-state supercapacitors of different compositions at a scan rate 200 mV/s.
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8 The nearly rectangular shape and enclosed area under the CV curves of all compositions indicate
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10 that the supercapacitor performance is composition dependent. The TG25 based supercapacitor
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12 shows the highest specific capacitance value because of the optimum composition of GO and
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14 TiO₂ provides good electrical contact and maximum surface area (see Table S2 ESI). Figure 5 (b)
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16 represents the CV curve of TG25 based supercapacitor at various scan rates (from 2 to 200
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18 mV/s). The CV curves of the TG25 based supercapacitor maintain nearly rectangular shapes
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20 without having a faradic current at all the scan rates, indicating its excellent capacitive behavior
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22 originating from the electrical double layer (EDL) formation. It may be noted that the bare TiO₂
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24 has a very low electrochemical capacitance (<40 $\mu\text{F cm}^{-2}$) because of its low conductivity and
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26 negligible faradic capacitance.[27]
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34 Galvanostatic charge-discharge (GCD) curves are shown in Figure 5 (c) and (d) for different
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36 compositions at 0.5 mA/cm² current density and different current densities of TG25 based
37
38 device, respectively. The charge-discharge curves are nearly triangle shape having a negligible
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40 internal resistance (IR) drop, indicating easy adsorption and desorption of ions. According to the
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42 GCD curves, the specific capacitances of the composites are calculated in the potential windows
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44 ranging from 0 to 1V. It is found that the TG25 based supercapacitor shows a highest specific
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46 capacitance of ~73.43 mF/cm². The specific capacitance value of the TG25 based supercapacitor
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48 is compared to that of other GO/RGO-metal oxide based solid-state supercapacitor and displayed
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50 in Table 1. The fabricated TG25 based device outperforms other devices, indicating the GO-TiO₂
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53 as a potential candidate for energy storage applications.
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Electrode Material	Method	Specific capacitance	Potential window	Cyclic stability/ Specific capacitance retention	reference
Multilayer hybrid films of graphene and titanium dioxide	Layer-by-layer (LBL) self-assembly technique	0.857 mF cm ⁻²	-0.1-0.7V	98% capacity retention after 1500 cycles.	[28]
rGO/TiO ₂ NR/rGO	Hydrothermal preparation	114.5 F g ⁻¹	0-0.8V	85% of its initial capacitance after 4000 cycles.	[29]
Carbon/TiO ₂ /reduced graphene oxide composite	Vapour-induced phase separation and self-assembly processes	23.6mF cm ⁻²	-0.2-0.8V	97.6 % maintenance of the initial capacity after 500 cycles	[2]
NF/rGO/H-Fe ₂ O ₃	Surfactant-free hydrothermal method	37.88 mF cm ⁻²	0-1.5V	97% of capacitance retention after 10000 cycles	[30]

MnO ₂ -deposited graphene fibers	Facile wet-spinning assembly method	59.2 mF cm ⁻²	0-1.6 V	92.7% initial capacitance retention after 8000 cycles	[31]
GO-TiO ₂	Solvothermal atmospheric pressure.	73.43mF cm ⁻²	0-1 V	92% capacity retention after 10000 cycles	This Work

Table 1: Comparative performance of GO/RGO-metal oxide based solid-state supercapacitors

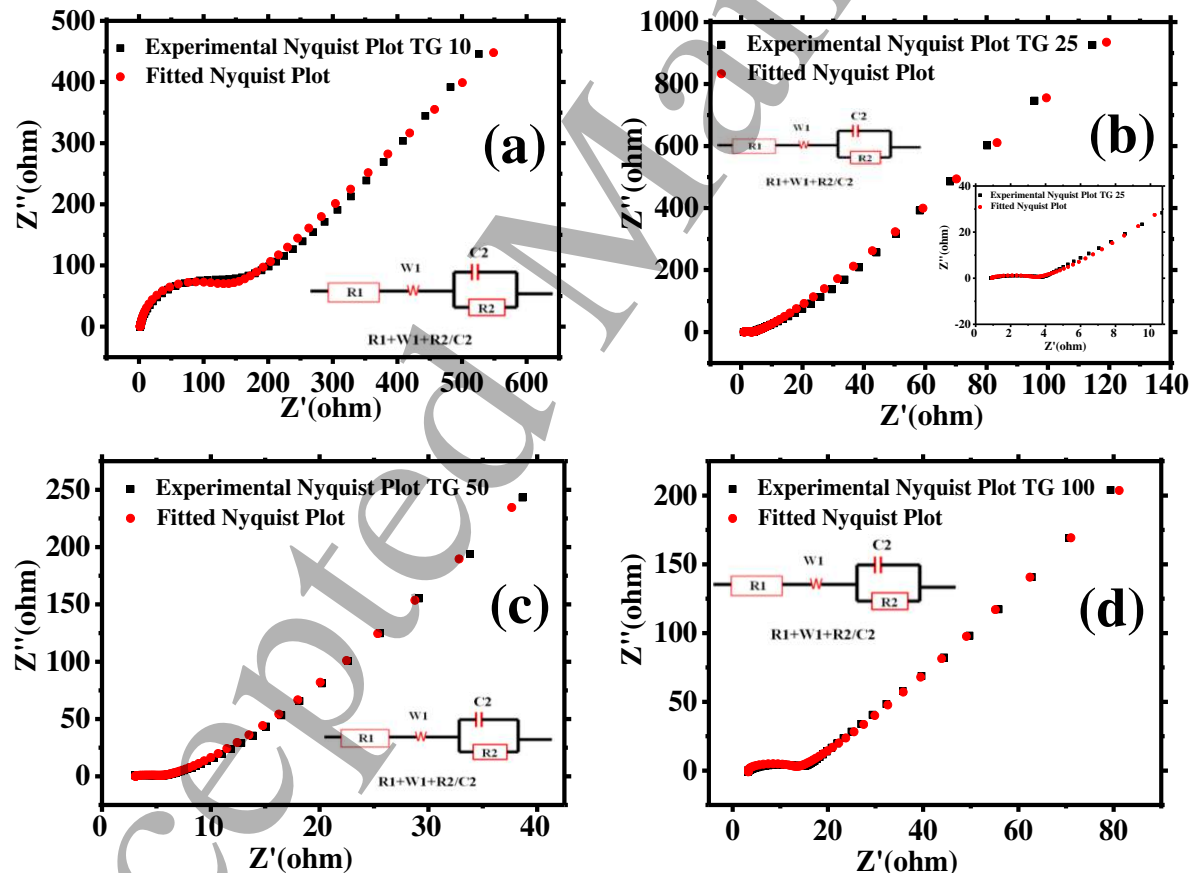


Figure 6: Typical Nyquist plots fitted with equivalent circuit diagram (inset) for different compositions based supercapacitor (a) TG 10, (b) TG 25, (c) TG 50 and (d) TG 100.

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3 To investigate the origin of high specific capacitance, electrochemical impedance measurements
4 were carried out over a frequency range from 0.001 Hz to 10 kHz. Figure 6 shows the Nyquist
5 plots of the all compositions of GO-TiO₂ nanocomposite based supercapacitors. Nyquist plot for
6 each composition shows a single semicircle and straight line slope in the high-frequency region
7 and low frequency regions, respectively. The sloping straight line in the low frequency region is
8 attributed to the ion diffusion/transportation in the electrode, and the single semicircle
9 corresponds to the interfacial charge-transfer resistance between electrodes and electrolyte.
10 Charge transfer resistance (R_{ct}) values of TG10, TG25, TG50, and TG100 based supercapacitors
11 are calculated to be 113.5, 2.3, 3.1, 9.3 ohm, respectively after fitting the Nyquist plot in the
12 equivalent circuit shown in the inset of Figure 6 a-d. It is found that R_{ct} value for TG25 based
13 supercapacitor is the lowest and that's the reason for high specific capacitance value of TG25
14 based supercapacitors.[32] The equivalent series resistance (ESR) values are found to be 1.64,
15 0.82, 2.30 and 3.17 Ω , deduced from the point of the curve intersection with a lateral axis for
16 TG10, TG25, TG50 and TG100, respectively. ESR is the sum of electrolyte resistance, intrinsic
17 resistance of the active materials and the contact resistance between electrode electrolyte
18 interfaces. The low value of ESR confirms a good contact between electrode materials and
19 current collector (steel) and a small resistance between the electrode and electrolyte interfaces.
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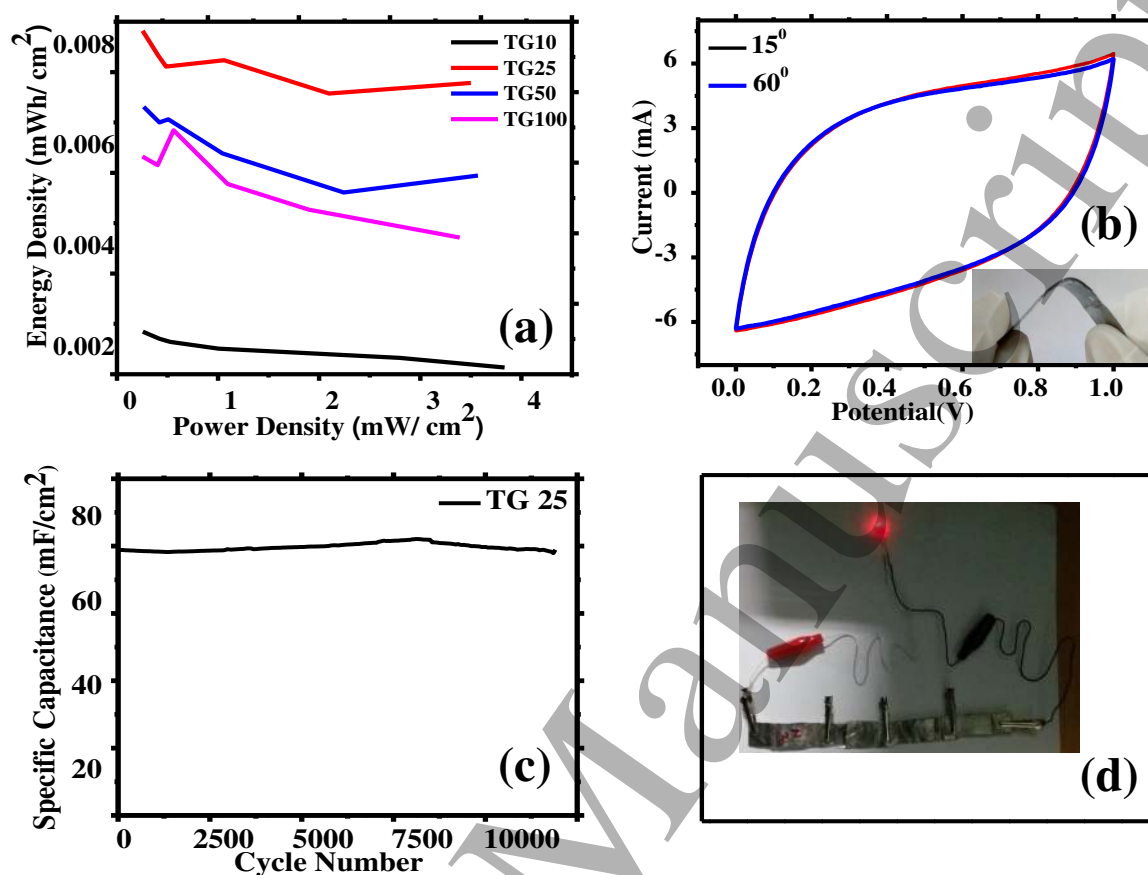


Figure 7: (a) Comparative Ragone plot of different compositions based device; (b) comparison of CV curves at different bending angles; (c) Cyclic stability of the TG25 based device; (d) LED illumination using fabricated supercapacitor.

Ragone plots shown in Figure 7(a) describe the corresponding relation between the specific energy density (E) and the specific power density (P), which are the main parameters that reflect the electrochemical performances of the supercapacitors. The energy densities of the TG25 supercapacitor were dramatically higher compared with that of other composites based supercapacitor, especially in cases with lower power densities (Table S3, ESI), indicating it as an ideal composition to get superior performances.

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3 The flexibility of the devices is also tested by measuring the CV curve at different bending
4 angles. The unaltered CV curves (Figure 7b) at different bending angles confirm the
5 nanocomposites as a potential candidate for flexible energy storage technology. Interestingly, the
6 TG25 based supercapacitor device shows high (92%) specific capacitance retention after
7 continuous 10,000 charge discharge cycles (7 c). Finally, to provide a clear evidence of the
8 electrical charge storage property, we have illuminated a red LED, as shown in Figure 7(d) and
9 video S1 in the ESI. A series combination of three supercapacitors is charged by applying 2V for
10 3 min. After disconnecting it from the voltage sources, it glows the red LED for 5 min and then it
11 starts discharging and fully discharged within next 5 min.
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24 **Conclusions:**

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28 In this article, mesoporous GO-TiO₂ nanocomposites consisting varying proportion of GO are
29 synthesized by a simple, easy, fast and controlled HCl catalyzed method for flexible solid-state
30 supercapacitor applications. Evolutions of TiO₂ nanospindles of rutile phase on graphene
31 platform are monitored by SEM, TEM, XPS, XRD and Raman analysis. Employing, a new
32 combination of GO-TiO₂ as electrode active layer material and PVA-H₃PO₄ as gel electrolyte, a
33 high performance solid-state supercapacitor is realized. The GO-TiO₂ composites based flexible
34 solid-state supercapacitor show a high specific capacitance of 73.43 mF/cm² at a current density
35 of 0.5 mA /cm² and 92% retention of specific capacitance after 10,000 continuous charge-
36 discharge cycles. The superior performance of TiO₂ nanospindles grafted GO based
37 supercapacitor (TG25) is originated from its large surface area of mesoporous structure of TG25
38 and intimate contact between graphene oxide and TiO₂. Finally, three supercapacitors in a series
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3 have illuminated a red LED, indicating the material as a potential candidate for energy storage
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16 chemistry for fruitful discussion on BET adsorption isotherm analysis.
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